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Optically Defined Multifunctional Patterning of Photosensitive Thin-Film Silica Mesophases

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Photosensitive films incorporating molecular photoacid generators compartmentalized within a silica-surfactant mesophase were prepared by an evaporation-induced self-assembly process. Ultraviolet exposure promoted localized acid-catalyzed siloxane condensation, which can be used for selective etching of unexposed regions; for "gray-scale" patterning of refractive index, pore size, surface area, and wetting behavior; and for optically defining a mesophase transformation (from hexagonal to tetragonal) within the film. The ability to optically define and continuously control both structure and function on the macro- and mesoscales is of interest for sensor arrays, nanoreactors, photonic and fluidic devices, and low-dielectric-constant films.

The ability to pattern porous thin films is important for a number of technological applications, including sensor arrays, optics, and microfluidic devices. Mesoporous silicas (1) are attractive for such applications because they have internal connectivity and variable density that can be tailored by preparative conditions. Soft lithographic approaches (2) have been used to pattern mesoporous films but require long processing times (3, 4) or have been limited to physically defining the presence or absence of dis-

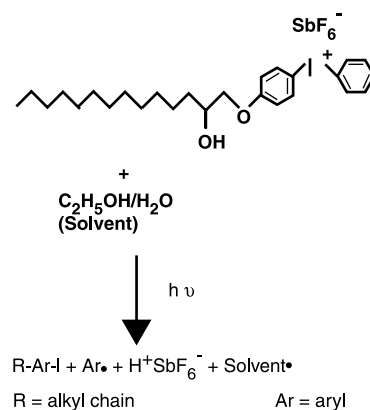
crete isolated regions (3, 4). Rapid patterning of organofunctionalized mesoporous thin films by means of pen lithography, ink-jet printing, and selective dewetting has also been demonstrated recently (5), but to date no one has successfully patterned thin-film mesostructure or properties.

Here we report patterning of the mesostructure within a thin film. Our procedure uses evaporation-induced self-assembly to prepare a photosensitive thin-film mesophase containing a photoacid generator (PAG). We then exploit the pH sensitivity of both the siloxane condensation rate and the silica-surfactant self-assembly process to optically define film location, mesostructure, and properties. The procedure begins with a homogeneous solution of silica, surfactant, PAG (a diaryliodonium salt), and HCl (6), with initial acid concentration designed to minimize the siloxane condensation rate (7, 8). Preferential ethanol evaporation during dip- or spin-coating (9) concentrates the depositing solution in water and nonvolatile constituents, thereby promoting self-assembly

(10, 11) into a photosensitive, one-dimensional hexagonal (1-dH) silica-surfactant mesophase. Because it bears a long-chain hydrocarbon, the PAG serves as a cosurfactant during the assembly process, which promotes its uniform incorporation within the mesostructured channels of the 1-dH film.

Irradiation of the PAG at a maximum wavelength (λ_{max}) of 256 nm (reaction 1) results in homolytic or heterolytic photodecomposition to yield the Brønsted superacid, H^+SbF_6^- , plus an iodoaromatic compound and organic byproducts (12). Thus ultraviolet (UV) exposure of the photosensitive mesophase through a mask creates patterned regions of differing acid concentrations compartmentalized within the silica mesophase (Fig. 1). Co-incorporation of a pH-sensitive dye (ethyl violet) allows direct imaging of these patterned regions as in Fig. 2A, where the yellow (exposed) and blue (masked) regions correspond to pH 0 and pH 2, respectively (13).

Suppression of the siloxane condensation rate during film deposition enables several modes of optically mediated patterning. Because acid generation promotes siloxane condensation, selective UV exposure results in patterned regions of more and less highly condensed silica (14). Differential extents of siloxane condensation result in turn in differential solubility, allowing selective etching of more weakly condensed regions in aqueous base (0.2



Reaction 1

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M NaOH) (scheme 1 of Fig. 1). An optical micrograph of a UV-exposed and etched thin-film mesophase after calcination to remove the

surfactant templates (Fig. 2B) reveals that the film is present only in the exposed regions. The plan-view transmission electron microscopy

(TEM) image (Fig. 2B, inset) reveals a striped mesoscopic structure consistent with a 1-dH mesophase with unit cell size $a = 37 \text{ \AA}$. We have also used a C_3F_8 plasma to directly etch the unexposed regions of photopatterned films (15). This process could potentially eliminate the need for a photoresist in the patterning of mesoporous low-dielectric-constant films needed by the microelectronics industry (16).

Organic by-products of the (patterned) PAG photodecomposition process render UV-exposed regions of the film hydrophobic (contact angle = 40°) relative to adjoining unexposed regions (contact angle $<10^\circ$). This differential wetting behavior is evident in Fig. 2D, where a patterned, square hydrophilic lattice serves to corral water into an array of monosized droplets. Such corrals can be used to selectively derivatize hydrophilic regions with aqueous-based reagents (17) and/or biomolecules.

A second aspect of our approach is the optical definition of film mesostructure and associated structure-related properties such as pore size, pore connectivity, and refractive index (Fig. 2C). Depending principally on the type and concentration of surfactant, we can produce two contrasting types of behavior. As depicted in schemes 2 and 3 of Fig. 1, if we heat the patterned film, we obtain adjoining regions of differing refractive index where the sign of the refractive index contrast Δn may be either positive ($n_{\text{irradiated}} > n_{\text{unirradiated}}$) or negative. Both types of behavior result from a greater extent of siloxane condensation in the UV-exposed regions. For positive Δn , the UV-exposed regions are still hexagonal but are denser than unexposed regions. We used a cantilever beam technique (18, 19) to measure the development of condensation-induced, thin-film, biaxial tensile stress and found that the differing extent of siloxane condensation was UV-dose-dependent. Therefore, by spatially varying the UV exposure using grayscale lithography, we can pattern continuous variations in the film refractive index and thickness, as well as in the structural properties of the 1-dH thin-film mesophase, such as unit cell dimension, pore size, and surface area (Table 1).

We have used our ability to pattern the refractive index, as evinced by the optical interference image shown in Fig. 2C, to create optical diffraction gratings (20). Exposure of a photosensitive film (21) with an excimer laser ($\lambda = 248 \text{ nm}$) through a phase mask resulted in a 543-nm grating structure (22) with $\Delta n \sim 0.025$ (scheme 2 of Fig. 1) after calcination to remove the surfactant templates. The measured Littrow angle of 34° is consistent with the mask period of 564 nm. Compared to gratings patterned in glass (23, 24) or liquid crystals (25), the ordered porosity of our gratings allows the diffracted intensity to be tunable by and sensitive to certain adsorbing molecules in the environment.

Fig. 1. Processing pathways for optically defined multifunctional patterning of thin-film silica mesophases. Conc., concentration; irr, irradiated; unirr, unirradiated.

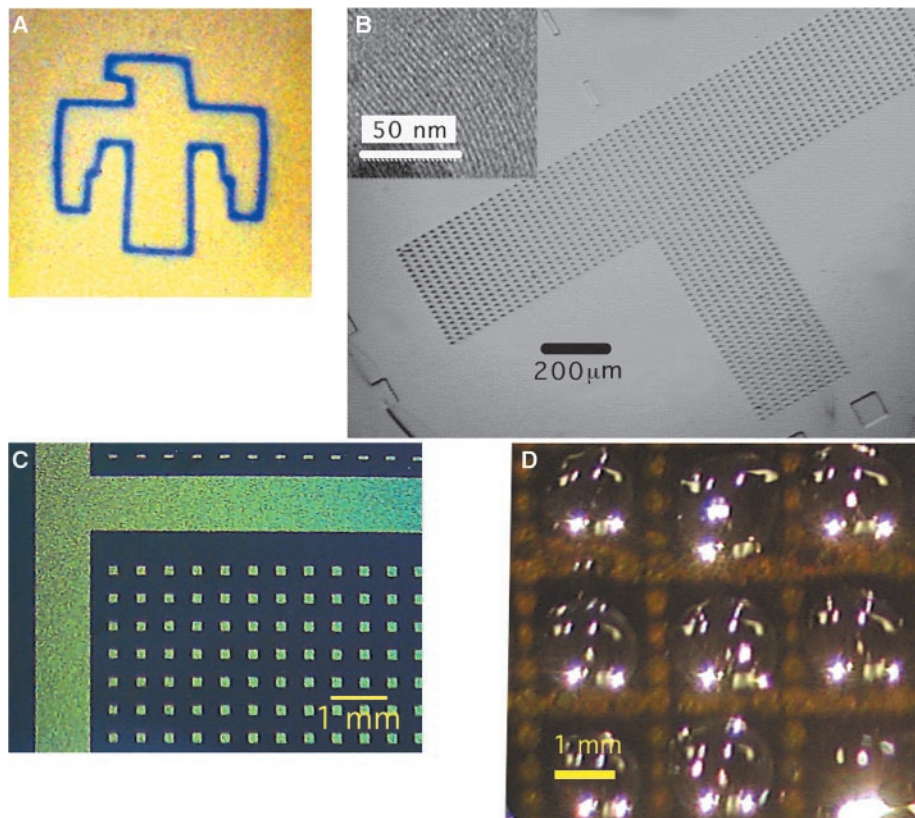
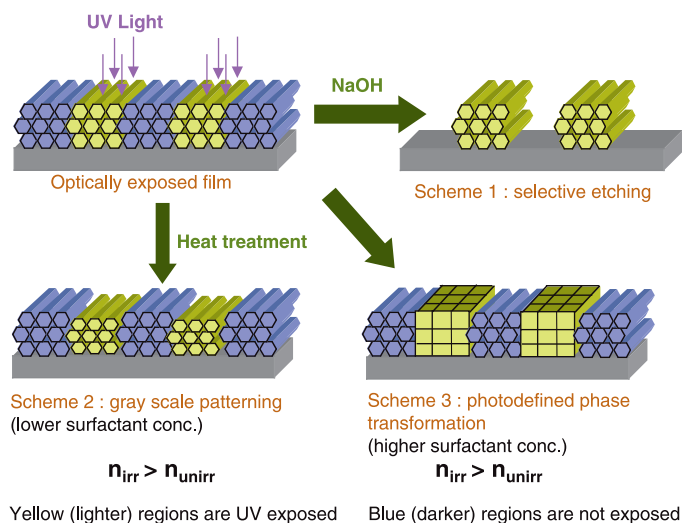


Fig. 2. Optical patterning of function and properties in thin-film silica mesophases. (A) Optical image of localized acid generation via co-incorporation of a pH-sensitive dye (ethyl violet). The blue areas observed on the unexposed film correspond to $\text{pH}^* \geq 2.0$, and the yellow areas observed on the exposed film correspond to $\text{pH}^* \sim 0$ (where pH^* refers to the equivalent aqueous solution pH required to achieve the same colors). (B) Optical micrograph of a UV-exposed and selectively etched mesostructured thin film (after calcination). Feature size $\approx 10 \mu\text{m}$. Inset: TEM image of the film seen in (B), consistent with the [110] orientation of a 1-dH mesophase with lattice constant $a = 37 \text{ \AA}$. (C) Optical interference image showing thickness and refractive index contrast in a patterned calcined film. The green areas correspond to UV-exposed and calcined regions and the black areas to unexposed and calcined regions. (D) Optical image of an array of water droplets contained within patterned hydrophilic-hydrophobic corrals. Water droplets sit on hydrophilic regions with contact angle $<10^\circ$ and are bounded by the hydrophobic UV-exposed regions with contact angle = 40° .

Using a higher initial surfactant concentration, near that required for the transformation from 1-dH to cubic without added acid, we observe a negative refractive index contrast that results from a hexagonal-to-tetragonal mesophase transformation (Table 2). Evidence for the optically defined phase transformation was obtained by x-ray diffraction (XRD) and TEM. After calcination at 450°C to remove surfactant, the XRD pattern of the unirradiated film (Fig. 3, trace A) is consistent with a 1-dH mesophase with lattice constant $a = 42.1 \text{ \AA}$, whereas the pattern of the irradiated sample (Fig. 3, trace B) is assigned to a tetragonal (distorted cubic) mesophase with $a = 66.8 \text{ \AA}$ and $b = c = 72.8 \text{ \AA}$ (26), along with residual traces of the untransformed parent structure (27). The XRD pattern of an unirradiated PAG-containing film exposed to HCl vapor (9) before calcination is comparable to that of the irradiated sample but shifted to a higher d -spacing (Fig. 3, trace C). The cross-sectional TEM image of the unirradiated film (Fig. 4A) reveals a striped pattern characteristic of the [110] orientation of a 1-dH mesophase ($a = 42.7 \text{ \AA}$), with mesopore channels oriented parallel to the substrate surface; that for the irradiated film (Fig. 4B) predominantly exhibits a diamond-shaped texture consistent with the [010] orientation of a tetragonal mesophase (28). The topotactic relationship between the 1-dH and tetragonal mesophases (Fig. 4, insets) indicates that the transformation results in a factor of $\sqrt{3}$ increase in lattice constant.

The mechanism of this phase transformation may be understood by considering how silica-surfactant mesophases reorganize in response to

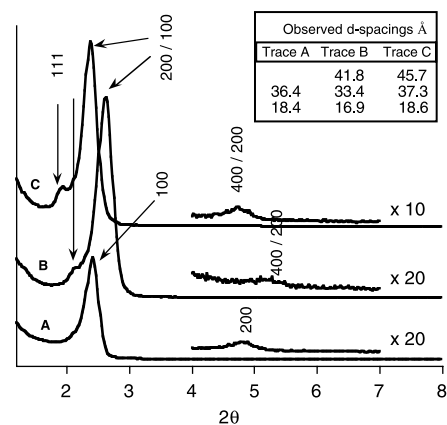


Fig. 3. X-ray diffraction patterns of calcined thin-film silica mesophases. Trace A: Unirradiated film pattern, consistent with a 1-dH mesophase with lattice constant $a = 42.1 \text{ \AA}$. Trace B: Irradiated film pattern, consistent with a tetragonal mesophase with $a = 66.8 \text{ \AA}$ and $b = c = 72.8 \text{ \AA}$, along with parent 1-dH mesophase with $a = 38.5 \text{ \AA}$. Trace C: An unirradiated film exposed to HCl vapors and then calcined. The pattern is similar to the one obtained for the irradiated film (trace B) but with lattice constants $a = 74.2 \text{ \AA}$ and $b = c = 82 \text{ \AA}$ for the tetragonal mesophase and $a = 43 \text{ \AA}$ for the 1-dH mesophase.

increased extents of siloxane condensation. The thermodynamically favored phase is that which allows the surfactant headgroup area a to be closest to its optimal value a_0 , while maintaining favorable packing of the hydrophobic surfactant tails (29, 30). The influence of a_0 along with the surfactant volume v and tail length l on the resultant mesophase can be understood qualitatively by the dimensionless packing parameter $g = v/a_0l$, where $g = 1$ favors the formation of vesicles, bilayers, or lamellar mesophases, and decreasing values of g result in the formation of progressively higher curvature mesophases and ultimately spherical micelles ($g < 1/3$) (29, 30). Monnier *et al.* (31) introduced the term G_{inter} in their free energy expression to account for electrostatic interactions between the silica framework and surfactant head groups. When the framework charge density matches the average surface charge density of the surfactant head groups $1/a$, G_{inter} is minimized, establishing a_0 , which in turn influences g and the mesophase curvature.

In the present study, condensation of the silica framework substantially increases its acidity, as reflected by a reduced $\text{p}K_a$ (where K_a is the acid dissociation constant) and isoelectric point (7). Therefore, although the overall pH of the system is reduced by photoinduced acid generation, the protonation of the silica framework decreases relative to that of the ethylene

oxide head groups. In order to maintain charge density matching at the silica-surfactant interface, the optimal ethylene oxide headgroup area a_0 must increase. The increased value of a_0 in turn reduces the surfactant packing parameter g , favoring transformation to a higher curvature mesophase. This transformation to a lower density mesophase is no doubt aided by the condensation-induced tensile stress, which in effect stretches the film, fostering its transformation to a lower density form.

Mechanistically, we propose that the phase transformation proceeds through the creation of periodic undulations along the length of the close-packed cylindrical channels of the 1-dH mesophase, as is known for the temperature-induced hexagonal-to-body-centered cubic transformation of $\text{C}_{12}\text{EO}_{12}$ in the $\text{C}_{12}\text{EO}_{12}/\text{H}_2\text{O}$ binary system (32). For the silica-surfactant system, undulation results in a tetragonal (distorted cubic) packing as depicted in Fig. 5. d_{100} for the parent 1-dH mesophase becomes d_{200} of the tetragonal mesophase, resulting in a factor of $\sqrt{3}$ increase in the lattice parameter. Distortion arises because of shrinkage that is normal to the substrate. The phase transformation from 1-dH to tetragonal occurs with minimal displacement of the silica oligomers and surfactant species, and we expect a precise topotactic relationship between the hexagonal and tetragonal mesophases as shown schematically in Fig.

Table 1. Properties of films resulting from differing UV exposure times followed by calcination at 425°C.

Exposure (min)	Thickness (nm)	n	d_{100} (Å)	Unit cell (Å)	Porosity*	Pore diameter† (Å)	Pore volume (V)* (cc/g)	Surface area (A)* (m ² /g)
0	192.5	1.318	44.58	51.44				
1	190.3	1.330	42.90	49.50	50.5%	19.0	0.38	813
2	189.4	1.333	41.85	48.29	47.3%	18.5	0.34	736
3	184.7	1.359	39.90	46.04	39.3%	17.9	0.24	548

*Determined by the surface acoustic wave (SAW)-based N_2 adsorption technique (35). The calcination temperature was reduced to 425°C for compatibility with SAW substrates. †Pore diameter = 4 V/A.

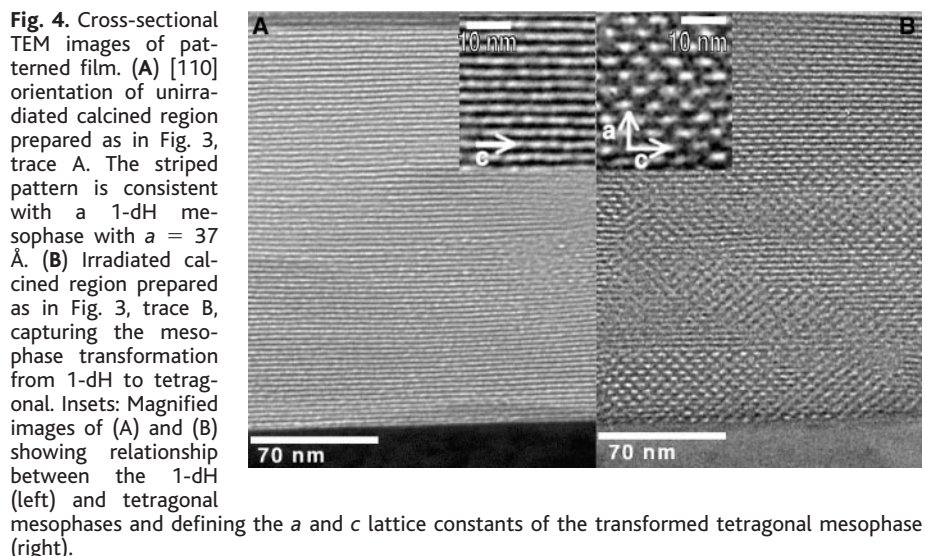


Fig. 4. Cross-sectional TEM images of patterned film. (A) [110] orientation of unirradiated calcined region prepared as in Fig. 3, trace A. The striped pattern is consistent with a 1-dH mesophase with $a = 37 \text{ \AA}$. (B) Irradiated calcined region prepared as in Fig. 3, trace B, capturing the mesophase transformation from 1-dH to tetragonal. Insets: Magnified images of (A) and (B) showing relationship between the 1-dH and tetragonal mesophases and defining the a and c lattice constants of the transformed tetragonal mesophase (right).

Fig. 5. Schematic diagram showing the mechanism of the transformation from the 1-dH mesophase (top) to a tetragonal (distorted cubic) mesophase (bottom). As condensation of the siloxane framework proceeds, the framework charge density decreases relative to that of the EO head groups. The corresponding increase in the optimal surfactant head-group area drives the mesophase transformation from 1-dH to tetragonal through a periodically undulating intermediate, as is known for the thermally driven transformation from 1-dH to body-centered cubic of $C_{12}EO_{12}$ in the $C_{12}EO_{12}/H_2O$ binary system (32).

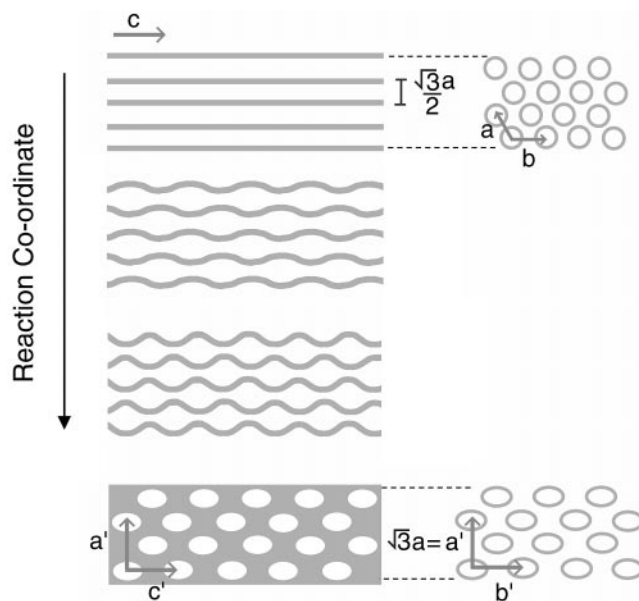


Table 2. Ellipsometry data showing thickness (t) and refractive index (n) (at $\lambda = 630$ nm) for films exhibiting the phase transformation from 1-dH to tetragonal.

	As-prepared films		Calcined films	
	t (nm)	n	t (nm)	n
Unirradiated	365.1 ± 2.6	1.454 ± 0.001	229.3 ± 2.2	1.302 ± 0.002
UV-irradiated	360.2 ± 2.1	1.457 ± 0.001	239.9 ± 2.3	1.277 ± 0.002

5 and by TEM in Fig. 4, A and B. There are two previous reports of condensation-driven transformations to higher curvature mesophases: lamellar to 1-dH (31) and lamellar to cubic (11), but our study appears to be the first report of a hexagonal-to-cubic or tetragonal transformation. Because it depends critically on the initial surfactant and acid concentrations (33), the transformation may be realized or avoided (scheme 3 versus scheme 2 of Fig. 1) by judicious choice of these parameters.

Although optical patterning of average properties, such as density, of disordered sol-gel films has been reported previously (23, 24, 34), the uniform initial pore size of our photosensitive thin-film mesophases, combined with the UV-dose dependence of our patterning procedure, provide a unique ability to precisely adjust and spatially define the pore size along with the refractive index and surface area by means of grayscale lithography. The observed hexagonal-to-tetragonal phase transformation further allows patterning of pore connectivity. These features are of practical importance for applications such as membranes and microfluidic systems.

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- Films were deposited on (100) silicon by dip-coating at 25.4 cm/min. Within 30 to 60 min of deposition, the samples were irradiated at 256 nm for times ranging from 15 s to 2 hours through a mask via proximity printing to effectively transfer the pattern onto the silica thin film. Etching of the films was done with a 0.2 M aqueous NaOH solution for a period of 5 to 10 s. Calcination treatments to promote continued siloxane condensation and remove surfactant templates and residual organics associated with PAG were conducted in air at 450°C for 3 hours, using a ramp rate of 1°C/min. HCl exposures of unirradiated films were done by placement in a chamber containing dispersed droplets of concentrated HCl for 2 hours.
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Vibrational Promotion of Electron Transfer

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By using laser methods to prepare specific quantum states of gas-phase nitric oxide molecules, we examined the role of vibrational motion in electron transfer to a molecule from a metal surface free from the complicating influence of solvation effects. The signature of the electron transfer process is a highly efficient multiquantum vibrational relaxation event, where the nitrogen oxide loses hundreds of kilojoules per mole of energy on a subpicosecond time scale. These results cannot be explained simply on the basis of Franck-Condon factors. The large-amplitude vibrational motion associated with molecules in high vibrational states strongly modulates the energetic driving force of the electron transfer reaction. These results show the importance of molecular vibration in promoting electron transfer reactions, a class of chemistry important to molecular electronics devices, solar energy conversion, and many biological processes.

The prototypical process of electron transfer is one of the most fundamental elementary chemical reactions found in nature. The pioneering work of Marcus (1) has exploded into a vigorous field of chemical and biochemical research, which has been recently reviewed (2). Electron transfer rates are controlled by two central dynamics: "solvent polarization" and "intramolecular vibrational motion." Under certain circumstances, it is the reorganization of the polarizable solvent about the moving charge that controls the rate of reaction (1). Under other circumstances, the electron transfer rate can only be understood by including effects of intramolecular vibration either classically (3) or quantum mechanically (4–6). Although hybrid models that incorporate both effects have recently been developed (7–11), our understanding of the dynamics of this important process is still far from complete. Furthermore, although studies of solvent influences on electron transfer have been quite successful (12, 13), the difficulties of performing state-resolved experiments in condensed phases have limited experiments on vibrational influences (14).

We found a way to at once remove the complicating effects of the solvent and to control the vibrational state of the reacting molecule by studying electron transfer dynamics between an incident gas molecule and a metal surface. This model system is in many ways

dynamically equivalent to condensed phase electron transfer, but solvent effects appear in a much simpler way. Specifically, the interaction of the ion with the solvent is replaced by an interaction with its image charge in the metal, which is much more easily modeled (15, 16). Furthermore, this approach allows the application of gas-phase optical pumping to prepare single quantum states in very high vibrational states in order to amplify the possible effects of vibration on electron transfer. We show that high levels of vibrational excitation can promote electron transfer from a metal surface to a gas-phase molecule. This work builds on previous condensed phase studies that have revealed substantial enhancements of electron transfer rates with vibrational excitation (14)

and provides a model system for the development of sophisticated charge transfer theories.

Electron transfer rates depend on molecular vibration because the shape of the molecule changes upon charge transfer. Classically, trajectories of vibrationally excited states can more easily traverse regions of configuration space similar to the stable anion. For the case of NO, we quantitatively calculated the form of the relevant molecular potentials (17) (Fig. 1A).

The difference between the two potential functions (Fig. 1B) is the vertical electron binding energy versus N–O separation. Near their potential minima, there is little energy difference between the neutral and anion curves, consistent with the low electron affinity associated with NO. For vibrationally excited states, however, near the outer turning point of the vibration state $\nu = 15$ (bond length $R_{\text{NO}} \sim 1.6$ Å), the vertical attachment energy to form NO^- is >200 kJ/mol. At the inner turning point, electron attachment is endoergic by about the same amount. This simple analysis shows that the energetic constraints controlling the ability of NO to accept (or, concomitantly, the ability of NO^- to donate) electrons depend strongly on internuclear separation.

Previously, Newsom derived potential parameters for neutral and anionic curves that describe the interaction of NO with Ag(111) (16). These are shown in Fig. 2, adapted to the case of Au(111). The anionic curve is calculated by using a Coulombic interaction between the anion and the image charge of the metal. The neutral curve reflects the weak physical interactions of NO with the noble metals. The asymptotic separation between the neutral and

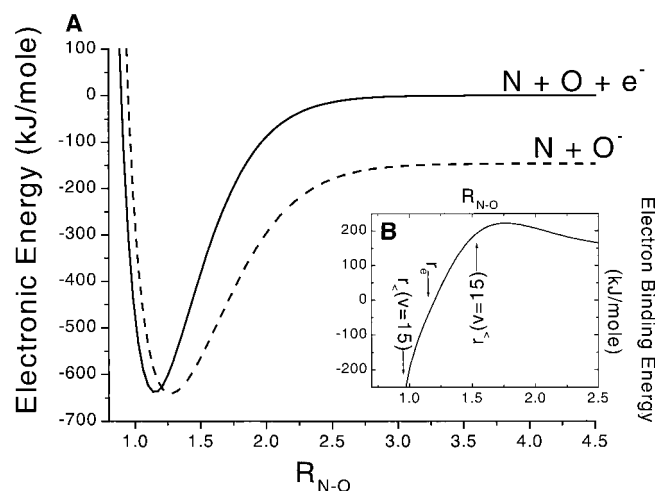


Fig. 1. (A) Ab initio calculations of NO and NO^- , showing the energetic constraints on electron transfers. (B) The difference between the two potential curves. The positions of the inner and outer extrema of the NO ($\nu = 15$) vibrational wave function are shown with vertical arrows labeled (r_e and r_v). The position of the NO potential minimum is shown as r_e .

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